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Nuclear magnetic resonance study of the enhanced ferromagnetic ordering in polycrystalline $SrRu_{1-x}Cr_xO_3$ ($0 \le x \le 0.12$)

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Abstract

The ferromagnetic ordering temperature of SrRu_{1-x}Cr_xO₃ increases to 175 and 186 K for x = 0.05 and 0.12, respectively, from 162 K for SrRuO₃ (x = 0). ⁵³Cr and ^{99,101}Ru nuclear magnetic resonance reveals that Cr is in a 'valence state' of Cr³⁺ ($t_{2g}^{3\uparrow}$), and Ru is in a mixed valence state, namely, Ru⁴⁺ ($t_{2g}^{3\uparrow1\downarrow}$) and Ru⁵⁺ ($t_{2g}^{3\uparrow}$). A single Ru NMR signal corresponding to Ru^{(4+\delta)+} is observed, indicating that the spin-down electron in the Ru 4d shell is less localized in SrRu_{1-x}Cr_xO₃ ($x \neq 0$). This result is consistent with a broadened Ru t_{2g} band and a possible Ru⁴⁺ (d⁴)–O²⁻–Ru⁵⁺ (d³) as well as Ru⁴⁺ (d⁴)–O²⁻–Cr³⁺ (d³) double-exchange interaction. This exchange interaction involves the Cr³⁺ in the ferromagnetic ordering and enhances the ordering temperature.

1. Introduction

SrRuO₃, the only known 4d transition metal oxide which is ferromagnetic, has attracted much attention due to its rich electronic and magnetic properties [1, 2] and possible technological applications [3]. SrRuO₃ orders ferromagnetically below $T_c \approx 160$ K and has an ordered moment as high as 1.6 μ_B/Ru [4]. Despite its robust ferromagnetism, the magnetic ordering temperature of SrRuO₃ can be easily suppressed [3–10]. A moderate reduction of the ordering temperature ($T_c = 140$ K) has been observed in thin films of SrRuO₃ deposited on (100) SrTiO₃ substrates [3]. This reduction is attributed to strain effects due to the lattice mismatch of 0.64%, which is consistent with the suppression of T_c under high pressure in bulk SrRuO₃ [5]. Under high pressure, T_c is reduced at a rate of $\partial T_c/\partial P = -5.7 \pm 0.2$ K GPa⁻¹ [5]. Annealing

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under a high-pressure (HiP) oxygen atmosphere introduces randomly distributed vacancies into the Ru sites while keeping the O content constant [6]. As a result, T_c drops drastically to 45 K with 9% Ru vacancies [6]. This suppression is attributed to an increased Ru formal valence. Substitution into Sr sites suppresses T_c as well. For example, isovalent Ca substitution for Sr suppresses T_c uniformly while maintaining the metallic conductivity [4, 7]; a small concentration of heterovalent Na ($\leq 12\%$) which is substituted for Sr reduces T_c rapidly to 70 K and induces an insulating state [8]. In the former case, the suppression is attributed to an increased orthorhombic distortion and a larger deviation of the Ru–O–Ru bond angle from 180° [2], while in the latter case, the suppression is thought to be similar to the high-pressure effect [8]. Zn²⁺, Ni²⁺, Co²⁺, and Mn³⁺ doped into the Ru sites decrease T_c significantly, down to 105 K [9]. This suppression of T_c is thought to be due to the interaction between the doping cations and the Ru t_{2g} band, which results in a narrowed bandwidth [9]. Finally, a low- T_c magnetically ordered phase of SrRuO₃ has been reported which apparently depends on the synthesis temperatures [10].

With all of the substitutions known so far, Cr^{3+} , when substituted for Ru, has been found to increase T_c up to 188 K and is the lone exception, making it very special [9]. A broadened Ru t_{2g} bandwidth has been suggested in this case [9]. In order to understand the microscopic origin of the enhancement of T_c , and how Cr is involved in the magnetic ordering, local studies using nuclear magnetic resonance (NMR), along with complementary magnetization measurements, have been carried out.

2. Experimental apparatus and procedure

Polycrystalline samples of stoichiometric SrRuO₃ and two Cr-substituted SrRu_{1-x}Cr_xO₃ compounds with x = 0.05 and 0.12 were synthesized from mixtures of SrCO₃, RuO₂ (prefired in air at 600 $^{\circ}$ C), and Cr₂O₃. Calcination of the starting mixtures was done for short periods of time at 800 °C to avoid the conspicuous volatility of RuO₂ at elevated temperatures. The intimately mixed powders were then pressed into pellets and fired in air at increasing temperatures with several intermediate grindings up to the sintering temperature of 1100 and 1370 °C for x = 0 and $x \neq 0$ samples, respectively. A final annealing was performed in air at 1100 and 1340 °C for the x = 0 and $x \neq 0$ samples, respectively. Three ≈ 2 g samples were obtained and examined by x-ray diffraction (XRD) on a Bruker powder diffractometer using Cu K α_1 radiation ($\lambda = 1.5406$ Å). X-ray diffraction patterns were recorded within $20^{\circ} \leq 2\theta \leq 80^{\circ}$ with a 2θ step of 0.02° . Within the sensitivity limit ($\approx 5\%$), all three samples show a single phase with the GdFeO₃-like orthorhombic structure. Magnetization measurements were carried out on a Quantum Design MPMS SQUID magnetometer for temperatures 5 K $\leq T \leq$ 350 K and magnetic fields -50 kOe $\leq H \leq$ +50 kOe. Zero-field spin-echo NMR was carried out on a conventional two-pulse spectrometer with phase sensitive detection. The NMR probe was located in a double dewar system with which temperatures down to 1.3 K were achieved by pumping on liquid helium. The NMR signals were optimized by adjusting the excitation power in a $\pi/2-\tau-\pi$ pulse sequence. Once NMR signals were optimized, the same excitation conditions were maintained over the entire frequency range of measurement. Usable spectra were obtained by averaging the NMR signals 500-2000 times at various frequency intervals across the spectrum.

3. Results and discussion

The temperature dependence of the DC magnetization presented in figure 1 clearly shows that the magnetic ordering temperatures of $SrRu_{1-x}Cr_xO_3$ ($x \neq 0$) are significantly increased



Figure 1. DC magnetization for $\text{SrRu}_{1-x}\text{Cr}_x\text{O}_3$ versus temperature in a field of 50 Oe upon cooling. T_c is increased to 175 and 186 K for samples with x = 0.05 and 0.12, respectively, from 162 K for SrRuO_3 .



Figure 2. Full hysteresis loops for SrRu_{1-x}Cr_xO₃ obtained at 5 K. Samples with $x \neq 0$ have smaller coercive fields than SrRuO₃ and the ordered moments per formula at 50 kOe are monotonically suppressed with x. The inset shows best fits to the Curie–Weiss law in the paramagnetic state.

with respect to SrRuO₃. The values of T_c are determined to be 175 and 186 K for samples with x = 0.05 and 0.12, respectively, compared with 162 K for SrRuO₃ (see table 1). These values are consistent with previous DC magnetization work [4, 9] as well as AC susceptibility measurements [11].

Table 1 summarizes all of the magnetic parameters obtained for $SrRu_{1-x}Cr_xO_3$ in this work. Ordered moments and coercive fields listed in columns three and four are derived from the complete hysteresis loops at 5 K (figure 2). A low temperature moment (component) value of 1.40 μ_B/Ru is achieved for SrRuO₃ in the ferromagnetically-ordered state at the largest applied field of 50 kOe, which is in good agreement with previous work [4]. Saturation is not reached under these conditions (5 K and 50 kOe), consistent with the fact that SrRuO₃

Table 1. Magnetic parameters for $SrRu_{1-x}Cr_xO_3$.

x	$T_{\rm c}$ (K)	$\mu_z \ (\mu_B)^a$	$H_{\rm c}~({\rm Oe})$	$\Theta\left(\mathbf{K}\right)$	$\mu_{\rm eff} (\mu_{\rm B})^{\rm b}$
0	162	1.40	2000	164	2.72
0.05	175	1.32	1200	179	2.49
0.12	186	1.15	1300	187	2.21

^a Ordered moment per formula at T = 5 K and H = 50 kOe from hysteresis loops.

^b Paramagnetic moment per formula from Curie-Weiss fits.

is an itinerant ferromagnet with large anisotropy [12]. A smaller ordered moment value of 1.32 $\mu_{\rm B}$ /formula is obtained for the sample with x = 0.05, and this value is further reduced to 1.15 $\mu_{\rm B}$ /formula with x = 0.12. The coercive field values for SrRu_{1-x}Cr_xO₃ ($x \neq 0$), which are determined to be ≈ 1200 Oe, are smaller than that of SrRuO₃, which is 2000 Oe. These results (both reduced ordered moments and reduced coercive fields) are not intuitively expected, since Cr³⁺ (the Cr³⁺ valence state is confirmed by ⁵³Cr NMR as is seen later), with spin S = 3/2, would have a larger ordered moment $\langle S \rangle = 3 \,\mu_{\rm B}/{\rm Cr}$ than Ru⁴⁺, whose ordered moment would be 2 $\mu_{\rm B}/{\rm Ru}$ with the low spin configuration S = 1. Experimentally, a value of $\langle S \rangle \approx 2.82 \pm 0.20 \ \mu_{\rm B}/{\rm Cr}^{3+}$ is deduced for CrCl₃ by neutron diffraction [13]. It must be noted that assigning of an integral valence in an itinerant system is not strictly valid, and in this work, any assignment of an integer to the valence states for Cr and Ru is taken in the same sense. As for the coercive field, disorder induced by doping is expected to enhance the pinning effect on the domain walls and, consequently, the coercive field would become larger [9, 14]. For example, high-pressure (HiP) oxygen annealing of SrRuO₃ introduces a small amount (a few per cent) of randomly distributed vacancies in the Ru network. As a result, the coercive field is increased to 4500 Oe [14]. These results, while somehow unexpected, do indicate a modified magnetism in $SrRu_{1-x}Cr_xO_3$, suggesting that the Cr dopants are participating in the magnetic ordering and not just acting as defects. Since the itinerant ferromagnetism in SrRuO3 originates from the narrow Ru t2g band and the bandwidth is crucial in determining the nature of the magnetism [4], any contribution to broaden the band is expected to enhance the ferromagnetism. Cr^{3+} , due to its $t_{2g}^3 e_g^0$ electronic configuration and close ionic size to Ru^{4+} (0.615 Å versus 0.62 Å), favours hybridization with the Ru⁴⁺ t_{2g} band, resulting in a broadened bandwidth [9]. As a result, the ferromagnetism in $SrRu_{1-x}Cr_xO_3$ is enhanced as is evidenced by the increased ordering temperature. Columns five and six list the parameters deduced from Curie–Weiss (CW) fits of the susceptibility over the temperature range 250 K $\leq T \leq$ 350 K (see the inset in figure 2). The Curie–Weiss law is given by

$$\chi(T) = \frac{N\mu_{\text{eff}}^2}{3k_{\text{B}}(T-\Theta)} + \chi_0,$$
(1)

where N is the concentration of the magnetic moments, μ_{eff} is the effective paramagnetic moment (magnitude), k_B is the Boltzmann constant, Θ is the CW temperature, and χ_0 is a temperature-independent term which reflects the core diamagnetism, Landau diamagnetism, and Pauli paramagnetism. The experimental data fit a single CW law although two types of moments exist for $x \neq 0$, and Θ values of 164, 179 and 187 K are obtained for samples with x = 0, 0.05 and 0.12, respectively. μ_{eff} is determined to be 2.72 μ_B/Ru for SrRuO₃, consistent with a valence of Ru⁴⁺ with S = 1. It is noteworthy that μ_{eff} is monotonically reduced to 2.49 μ_B/f ormula and 2.21 μ_B/f ormula for samples with x = 0.05 and 0.12, respectively. These values are significantly smaller than the theoretical combination of Ru⁴⁺ (2.83 μ_B/Ru with S = 1) and Cr³⁺ (3.87 μ_B/Cr with S = 3/2). In a study of the Sr_{1-x}Ca_xRuO₃ system, the paramagnetic moments of Ru were reported to increase somewhat with x, while the Ru



Figure 3. Zero-field spin-echo NMR spectra for $SrRu_{1-x}Cr_xO_3$ at 1.3 K. (a) x = 0, the peaks at 64.4 and 72.2 MHz are due to ⁹⁹Ru and ¹⁰¹Ru isotopes, respectively. (b) x = 0.05, the sharp peak centred at 60.8 MHz is attributed to ⁵³Cr and the two peaks at 65.3 and 73.3 MHz are attributed to ⁹⁹Ru and ¹⁰¹Ru, respectively. (c) x = 0.12, the sharp peak centred at 60.3 MHz is attributed to ⁵³Cr, and the broad spectrum (\bullet) between 62 and 90 MHz is attributed to ^{99,101}Ru. Open circles (O) show the spectrum after subtraction of the intensity from the SrRuO₃ impurity phase and the line is a fit to the data using Gaussian peaks indicated by the arrows (see text).

 t_{2g} bandwidth decreases with x [15]. Thus, the small paramagnetic moments obtained in SrRu_{1-x}Cr_xO₃ ($x \neq 0$) may also hint at a broadened t_{2g} band due to the participation of the Cr³⁺ in the band formation.

Figure 3 shows the zero-field spin-echo NMR spectra obtained at 1.3 K for $SrRu_{1-x}Cr_xO_3$. The NMR spin-echo signals for the two samples with $x \neq 0$ are considerably weaker than that of $SrRuO_3$. In order to compare the spin-echo amplitudes directly, a factor (by which the amplitude has been multiplied) is provided in figure 3. The NMR spectrum for $SrRuO_3$ (figure 3(a)) consists of two well-defined peaks at 64.4 and 72.2 MHz, assigned to the ⁹⁹Ru (gyromagnetic ratio $\gamma = 0.19645$ MHz kOe⁻¹, nuclear spin I = 5/2) and ¹⁰¹Ru ($\gamma = 0.22018$ MHz kOe⁻¹, I = 5/2) isotopes, respectively. These NMR peak frequencies correspond to a hyperfine field of -328 kOe at the Ru nuclei, which is characteristic of the low-spin state of Ru⁴⁺ with S = 1 [15]. The NMR spectra for the two samples with $x \neq 0$ can be considered to fall within two frequency ranges. For the sample with x = 0.05 (figure 3(b)), the sharp peak centred at 60.8 MHz is attributed to ⁵³Cr and the two peaks located between 62 and 82 MHz are attributed to ^{99,101}Ru. Similarly, for the sample with x = 0.12 (figure 3(c)), the peak centred at 60.3 MHz is assigned to ⁵³Cr and the broad spectrum with features located between 62 and 90 MHz is assigned to ^{99,101}Ru.

Because of the 53 Cr NMR peak frequencies, Cr is believed to be in a valence state of Cr³⁺. In a series of chromium trihalides, where Cr is in the Cr³⁺ state [13], the 53 Cr NMR

Table 2. NMR parameters for $SrRu_{1-x}Cr_xO_3$.

x	$H_{\rm hf}^{\rm Cr}$ (kOe)	$H_{\rm hf}^{\rm Ru}$ (kOe)	$^{101}\nu_Q^{Ru}~(MHz)$	$^{99}\nu_Q^{Ru}~(MHz)$
0	_	-328	0.44 ^a	0.077 ^a
0.05	-253	-333	1.6	0.28
0.12	-250	-348	4.7	0.81

^a Data from [15].

peaks were found within 50–63 MHz [16]. With a reasonable assumption that the hyperfine coupling constant of Cr is independent of the valence state, frequency ranges of 67-84 and 33-42 MHz are estimated for ⁵³Cr²⁺ and ⁵³Cr⁴⁺, respectively. A search for ⁵³Cr⁴⁺ was made over the frequency range 27-50 MHz; however, no detectable signal was found. Given the facts that Cr³⁺ is the most stable valence state for chromium which forms stable salts with all the common anions and that Cr⁴⁺ is largely confined to peroxo- and fluoro-complexes [17], the peaks found at 60.8 MHz (x = 0.05) and 60.3 MHz (x = 0.12) can be taken as the evidence that Cr is in the Cr³⁺ state for SrRu_{1-x}Cr_xO₃. For ⁵³Cr, $\gamma = 0.2406$ MHz kOe⁻¹ and I = 3/2, which result in hyperfine fields (H_{hf}^{Cr}) of -253 and -250 kOe for samples with x = 0.05 and 0.12, respectively. The negative value indicates a dominant core-polarization contribution to $H_{\rm hf}^{\rm Cr}$, which is confirmed by the negative shift of the NMR frequency in an external magnetic field. Typically, in a ferromagnetically-ordered system, the zero-field NMR signals are strengthened due to an enhancement of the applied rf field (H_1) . This enhancement originates from the coherent motion of the ferromagnetic moments with H_1 and makes the NMR signals observable with small amplitude of H_1 . The appearance of the ⁵³Cr NMR signal here is facilitated by the H_1 enhancement, and thus provides evidence that Cr is involved in the ferromagnetic ordering for $SrRu_{1-x}Cr_xO_3$. Due to the small fraction of Cr in these samples as well as the low natural abundance for 53 Cr (9.5%), the 53 Cr NMR signals are weak compared with the 99,101 Ru signals. In figure 3, the heights of the 53 Cr peaks have been multiplied by different factors for clarity.

The ^{99,101}Ru NMR peaks are progressively broadened with x, which is believed to be quadrupolar in origin as a consequence of the disorder introduced by Cr doping. The quadrupole effect, which originates from the interaction between the electric field gradient (EFG) and the nuclear quadrupole moment Q, is characterized by a quadrupole frequency v_Q given by

$$\nu_{\rm Q} = \frac{3e^2 q \, Q}{2I(2I+1)},\tag{2}$$

where eq is the EFG component V_{zz} , and I is the nuclear spin. An accurate determination of v_Q from the broadening, however, is not possible. In an earlier work on SrRuO₃, quadrupole oscillations of the spin-echo amplitude were observed as a function of the separation τ between the two rf pulses, and v_Q was accordingly determined as 0.077 and 0.44 MHz for ⁹⁹Ru and ¹⁰¹Ru, respectively. Since the broadening is directly proportional to v_Q according to first-order perturbation theory, values of v_Q are estimated for SrRu_{1-x}Cr_xO₃ with $x \neq 0$ by comparing the FWHM width of the corresponding NMR peaks with that of SrRuO₃. As is seen in table 2, v_Q increases rapidly with x. Consequently, the ^{99,101}Ru peaks are no longer resolvable in the case of x = 0.12, and a broad spectrum is observed. Besides the broadening, the ^{99,101}Ru NMR peaks shift to higher frequencies with Cr doping. This shift, which is attributed to an increase of the hyperfine field H_{hf}^{Ru} (magnitude) at the Ru nuclei, indicates a change of the Ru valence state compared to that for the undoped SrRuO₃. Since Cr is doped in as Cr³⁺, charge neutrality requires that an equal amount of Ru⁴⁺ increases to Ru⁵⁺ by losing its spin-down electron in the

4d shell. The existence of some Ru^{5+} has also been suggested for the $SrRu_{1-\nu}O_3$ system, where v is the portion of vacancies in the Ru network introduced by either HiP oxygen annealing or a nonstoichiometric synthesis condition [6]. Also, Ru⁵⁺ is found to coexist with Ru⁴⁺ in the magnetic superconductor $RuSr_2GdCu_2O_8$ [18]. In this work, a search for Ru^{5+} was made over the frequency range 110-145 MHz, which is a typical range for Ru⁵⁺ [18]; however, no detectable NMR signal was found. As mentioned above, by the charge neutrality requirement, 5% and 12% of Ru⁵⁺ would be introduced into the samples with x = 0.05 and 0.12, respectively. By assuming that the enhancement factors for the Ru⁴⁺ and Ru⁵⁺ are essentially the same, and including the favourable factor of ≈ 2 for Ru⁵⁺ due to the higher frequency, an estimate for the Ru⁵⁺ NMR signal to noise can be made. Based on these considerations, the NMR spectrum for static Ru^{5+} would have approximately 11% and 27% of the intensity of Ru^{4+} for samples with x = 0.05 and 0.12, respectively, which is easily within the sensitivity of the instrumentation. The absence of the Ru⁵⁺ NMR signal indicates that there exists essentially no static Ru⁵⁺ in the magnetically-ordered state on a timescale longer than 10^{-8} s. Thus, it is likely that the spin-down electron in the Ru 4d shell is more itinerant in $SrRu_{1-x}Cr_xO_3$ ($x \neq 0$) compared with SrRuO₃ and, consequently, an average valence of Ru^{$(4+\delta)+$} results. This conclusion is consistent with the hypothesis that the Ru t_{2g} bandwidth is broadened with Cr doping [9] and creates the possibility of a $Ru^{4+}(d^4)-O^{2-}-Ru^{5+}(d^3)$ double-exchange (DE) interaction. This DE interaction, which may involve Cr^{3+} in the magnetic ordering as well, since Cr^{3+} has a similar electronic structure $(t_{2g}^{3\uparrow})$ to that of Ru⁵⁺ $(t_{2g}^{3\uparrow})$, enhances the magnetic ordering and increases the ordering temperature. The slight difference in ⁵³Cr NMR peak frequency for the two Cr-doped samples may be due to this DE exchange, with the spin-down electron having slightly larger probability on the Cr sites for the sample with x = 0.12. In a recent work on Mn-doped SrRuO₃, a similar DE interaction has been suggested [19]. δ , the average change in the Ru valence, is expected to be small due to the small fraction of Ru⁵⁺, which is consistent with the small shift in the 99,101 Ru NMR peak frequencies of SrRu_{1-x}Cr_xO₃ ($x \neq 0$) relative to that of SrRuO₃. From the 99,101 Ru NMR peak frequencies, $H_{hf}^{Ru} = -333$ kOe is determined for the sample with x = 0.05. The determination of $H_{\rm hf}^{\rm Ru}$ in the case of x = 0.12 is not so straightforward, since no distinct peaks are observed due to the much larger quadrupole effect. As is shown by solid circles in figure 3(c), besides the ⁵³Cr sharp peak and the ^{99,101}Ru main peak which are intrinsic to the x = 0.12 phase, some intensity is also found between 62 and 73 MHz. This intensity is believed to originate from a trace amount of SrRuO₃ second phase which is not detected by the XRD. A small peak at ≈162 K appears in the temperature derivative of the field-cooled magnetization, dM/dT, indicating the existence of SrRuO₃ impurity. The occurrence of SrRuO₃ second phase is almost inevitable in the preparation of polycrystalline samples involving Sr and Ru simultaneously [20]. It must be noted that the NMR signal from $SrRuO_3$ is particularly strong due to a favourable enhancement factor, and that even a minute amount can contribute signal intensity [15]. Subtracting a weighted SrRuO₃ spectrum (figure 3(a)) results in a corrected spectrum for the x = 0.12 phase which is shown by open circles in figure 3(c). A fit is made to this spectrum and is shown by the curve in figure 3(c). In making this fit, a Gaussian peak centred at 68.3 MHz is used for ⁹⁹Ru and a much broader peak centred at 76.5 MHz is used for ¹⁰¹Ru, which is the superposition of five peaks indicated by arrows. The reader is referred to Kumagai et al [18] for details of quadrupole splitting of the 99,101 Ru NMR spectrum. From this fit, a value of $H_{\rm hf}^{\rm Ru} = -348$ kOe is deduced for the sample with x = 0.12. Interestingly, if the angle between the Ru moment and the principal axis of the EFG tensor remains the same as that of SrRuO₃, a quadrupole frequency of 5.5 MHz is derived for ¹⁰¹Ru from this fit. As discussed above, a value of 4.7 MHz is derived by comparing the broadening of the NMR peaks with SrRuO₃. The difference is most likely due to the change in EFG as a result of the disorder introduced by Cr doping.

4. Conclusions

In conclusion, the ferromagnetic ordering temperature is increased with Cr doping into the Ru sites in SrRuO₃. The ⁵³Cr NMR results in a valence state of Cr³⁺. The ^{99,101}Ru NMR peaks are progressively broadened and shift to higher frequencies with Cr doping. The broadening is attributed to be quadrupolar in origin as a result of the disorder introduced by Cr doping. The shift in frequency indicates a change in the Ru valence state, i.e., part of the Ru⁴⁺ is increased to Ru⁵⁺. However, only an average valence of Ru^{(4+δ)+} is observed by NMR, which indicates that the spin-down electron in the Ru 4d shell is less localized. This result creates a possible Ru⁴⁺(d⁴)–O^{2–}–Ru⁵⁺(d³) as well as Ru⁴⁺(d⁴)–O^{2–}–Cr³⁺(d³) DE interaction which involves the Cr³⁺ in the magnetic ordering and enhances the ordering temperature.

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